

HPC SCHOOL ON QUANTUM COMPUTATIONAL MATERIALS SCIENCE



A. Marco Saitta

Institut de Minéralogie, de Physique des Matériaux et de Cosmochimie – Sorbonne U, CNRS, MNHN

Introduction to Computational Materials Science in Sorbonne: from DFT to machine learning methods

Paris/Singapore, Nov 17th 2020



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Who am I?

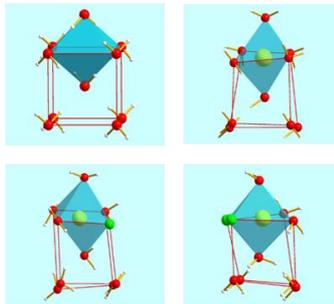
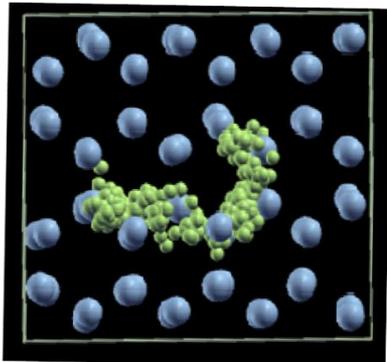
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 - Exotic properties of molecular crystals

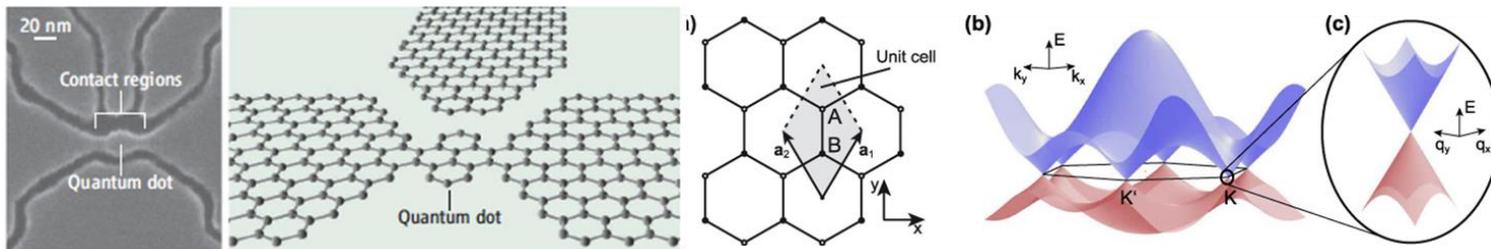


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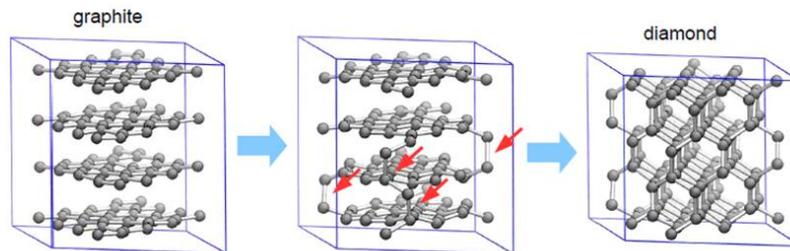
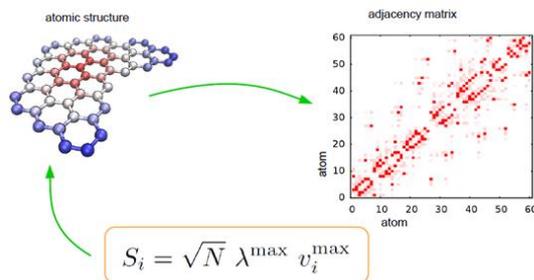
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 - Topology-based theory of transformations of matter (MAESTRO)



MAESTRO

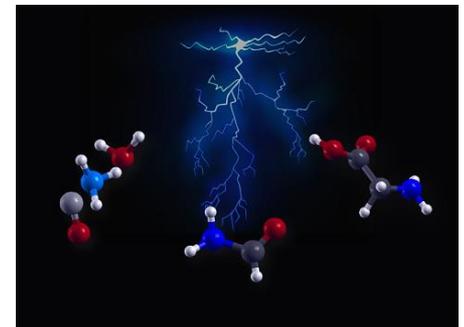


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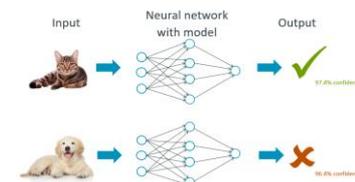
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 - Physical origins of biological life (NASA)



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 - Physical origins of biological life (NASA)
 - Machine-learning methods in computational materials science





Paris International School on Advanced Computational Materials Science

PISACMS

A. Marco SAITTA – Rodolphe VUILLEUMIER

Why

- Sorbonne: a singular generation of computational physicists and chemists working together
- Method development over the whole scale: from hydrodynamics models to Quantum MonteCarlo
- Strong support from our institutions; availability of computer labs
- Share our knowledge and pass it to new generations of computational physicists and chemists; help build new connections





What

- Classical Molecular Dynamics
- Density Functional Theory
- Ab Initio Molecular Dynamics
- Advanced Coarse-Graining and Mesoscale Methods
- Metadynamics & Free Energy Methods
- Density Functional Perturbation Theory
- Path-Integral Methods
- Classical & Quantum Monte-Carlo
- **Machine Learning methods**



Who

- Michele CASULA – IMPMC (Physics)
- Marie JARDAT – PHENIX (Chemistry)
- Maximilien LEVESQUE – PASTEUR (Chemistry)
- Michele LAZZERI – IMPMC (Physics)
- Virginie MARRY – PHENIX (Chemistry)
- Fabio PIETRUCCHI – IMPMC (Physics)
- Benjamin ROTENBERG – PHENIX (Chemistry)
- A. Marco SAITTA – IMPMC (Physics)
- Mathieu SALANNE – PHENIX (Chemistry)
- Ari SEITSONEN – PASTEUR (Chemistry)
- Rodolphe VUILLEUMIER – PASTEUR (Chemistry)



When & where

- Paris 2015: Aug 24 – Aug 28, UPMC Campus
- Paris 2016: Aug 29 – Sept 2, UPMC Campus
- Paris 2017: Aug 28 – Sept 1, UPMC Campus
- Bangalore 2018: joint SU-JNCASR « SJSACMS »
- Paris 2018: Aug 27 – Aug 31, SU PMC Campus
- Paris 2019: Aug 26 – Aug 30, SU PMC Campus
- **Paris 2020: Aug 24 – Aug 28, SU PMC Campus**
- **Singapore 2020: Nov 16 – Nov 20, NUS Campus**
- **Paris 2021**
- **Singapore 2021**
- **Abu Dhabi 2022 ?**
- **Bangalore 2022 ?**

<http://pisacms.sciencesconf.org>

MAterials for Energy through STochastic sampling and high peRformance cOmputing

MAESTRO

Team members

- Michele CASULA – IMPMC (Physics)
- Julien BIGOT – CEA (Computer Science)
- Ludovic GOUDENEGE – CentraleSupélec (Mathematics)
- Guillaume JOSLIN – CentraleSupélec (Computer Science)
- Pierre MONMARCHE – LJLL/LCT (Mathematics/Chemistry)
- Fabio PIETRUCCI – IMPMC (Physics)
- Benjamin ROTENBERG – PHENIX (Chemistry)
- **A. Marco SAITTA – IMPMC (Physics)**
- Mathieu SALANNE – PHENIX (Chemistry)
- Rodolphe VUILLEUMIER – PASTEUR (Chemistry)

Main Research Lines

- **Maths/Physics (WP1)**
 - Dimensionality reduction/coordinate optimization

- **Physics/Chemistry (WP2)**
 - (Energy-storage-interesting) ionic salts in solution

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A general overview of our scientific interests (with Fabio Pietrucci)

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 - Phase transitions (under pressure)
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 - Chemical reactions in solutions
 - Protein structure-function relations
- ❑ How: computer simulations of *ab initio* (= quantum) molecular dynamics
- ❑ One emblematic case: CO₂ at “geological” conditions
- ❑ Bottleneck: efficient sampling of 3N-dimensional configurational space
- ❑ Challenge n.1 : find “good” collective variables to describe/project the manifold onto a few relevant dimensions
- ❑ Challenge n.2 : replace costly *ab initio* simulations by machine-learning-based ones

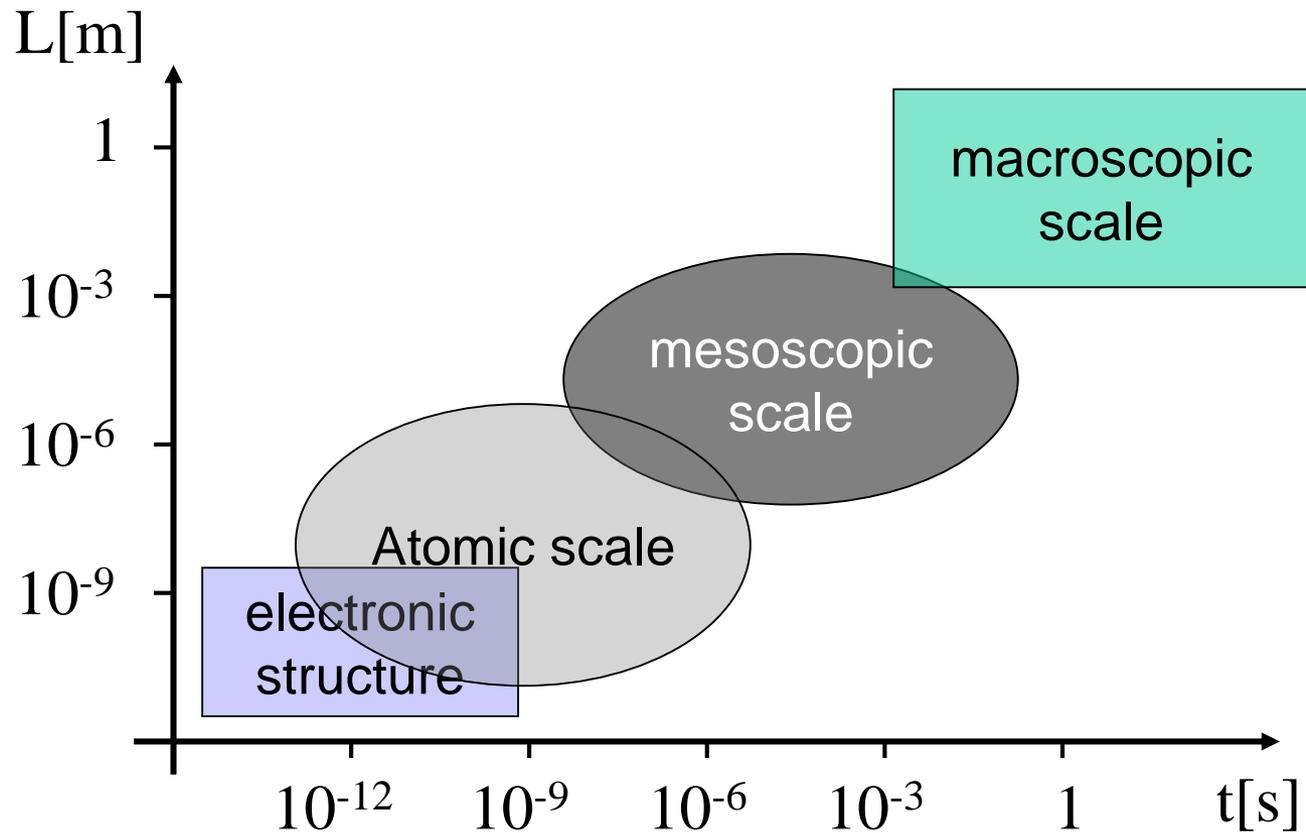
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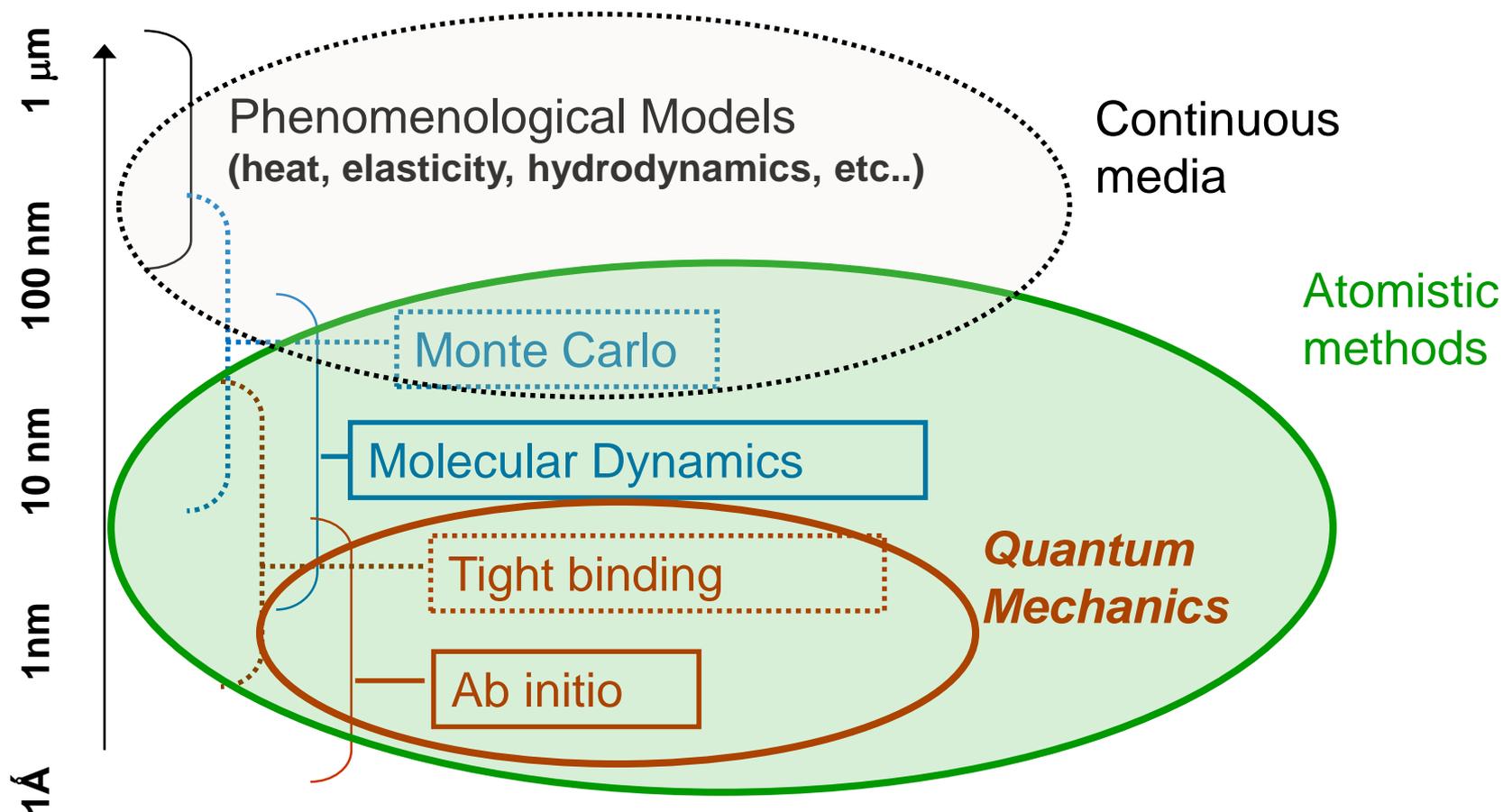
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Length/time scales, methods



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Length/time scales, methods



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Length/time scales, methods

Structural & dynamical properties
(from atomic trajectories)

Electronic & magnetic properties
(from electronic structure)

Increasing complexity

Empirical interatomic potentials

Quantum Mechanics

Tight-binding (semi-empirical)

Ab initio

DFT

WF based

Hartree-Fock

Quantum Monte Carlo

Post Hartree-Fock (MP2, Coupled Cluster, etc.)



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Ab initio DFT calculations

- *Ab initio* interactions solely determined by the electronic structure of atoms
- Solution of the quantum mechanical EOM of the electrons: energy and forces
- Atomic number the only “external” input: transferability
- However, a special theoretical framework is needed

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Ab initio DFT calculations

- Same approach from biology to planet interiors
- Electronic structure and properties, chemical reactions, change of coordination
- Accuracy at the (sub)nano scale
- Directly simulate experimental spectra (x-ray, Raman, infrared, NMR,...)

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Ab initio DFT calculations

Generality (« transferability »): all atoms from periodic table

Desired accuracy (« predictive » theory):

- crystallographic positions $\pm 0.005 \text{ \AA}$
- bonding energies $\pm 40 \text{ meV/atom}$ ($\cong 1 \text{ kcal/mol}$)
- vibrational frequencies $100 \text{ GHz} \sim 3.3 \text{ cm}^{-1}$
- dipolar moments $\pm 0.02 \text{ D}$
- magnetic moments $\pm 0.1 \mu_{\text{B}}$
- spectra (visible, UV, X) $\pm 0.1 \text{ eV}$

Typical system size:

$< 10^3$ atoms

Typical time scale:

$\sim 10^{-11} \text{ s}$

Computational efficiency:

parallel algorithms, modularity

The many-electron problem

Time-independent, non-relativistic, spinless, Schrödinger equation for N electrons interacting with fixed nuclei

$$H\Psi(\vec{r}_1, \dots, \vec{r}_N) = \varepsilon\Psi(\vec{r}_1, \dots, \vec{r}_N)$$

$$H = \sum_{i=1}^N \left[-\frac{\hbar^2 \nabla_i^2}{2m} + V_{\text{ext}}(\vec{r}_i) \right] + \sum_{i>j} \frac{e^2}{|\vec{r}_i - \vec{r}_j|}$$

sum of N one-electron operators
(kinetic operator + Coulomb
potential from the nuclei)

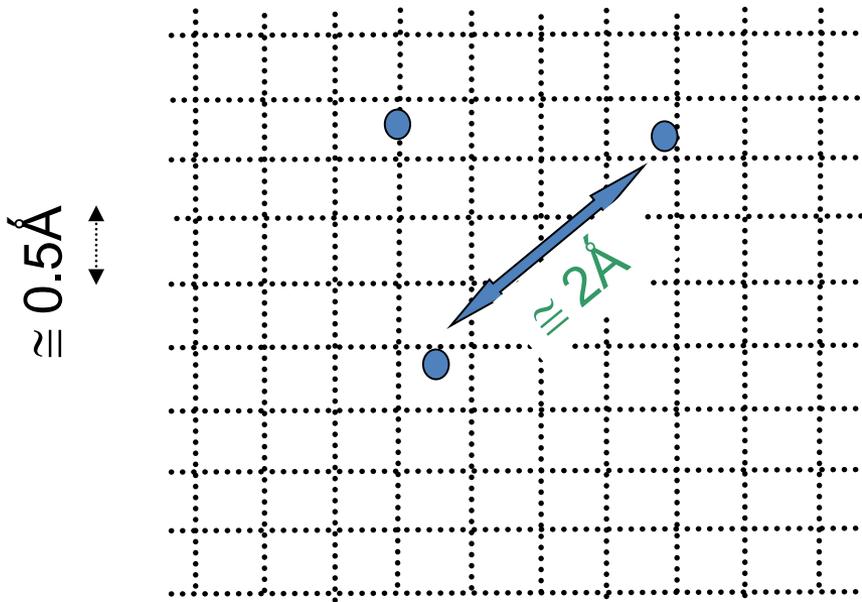
Long range $e^- - e^-$ interaction
(the big problem!)

Can we solve it numerically by brute force?

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The many-electron problem by brute force?

Example: Li_3 cluster



9 electrons

Solve the Schrödinger equation on a $M \times M \times M$ grid for each spatial coordinate

For $M=10 \rightarrow 10^{27} \times 2^9$

independent variables

$\rightarrow \sim 5 \times 10^{29}$ operations just to evaluate $\langle \Psi | H | \Psi \rangle$!

On a Tflops computer this would need $\sim 10^{11}$ years!

(unfeasible even if computer speed increases by a factor ~ 4 every ~ 2 yrs)

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Adiabatic (Born-Oppenheimer) approximation

- Electrons much lighter & faster than nuclei, their timescales much shorter
- Quantum EOM for electrons, classical ones for ions
- Solution of the electrons QM: global many-body wavefunction, whose size grows exponentially

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The electronic density

$$n(\vec{r}) = \langle \Psi | \sum_{i=1}^N \delta(\vec{r} - \vec{r}_i) | \Psi \rangle = N \int d^3r_2 \dots \int d^3r_N |\Psi(\vec{r}, \vec{r}_2, \dots, \vec{r}_N)|^2$$

- Is a function of 3 spatial coordinates, independently of N
- Is measurable
- Many physical quantities can be inferred from it

Can we reformulate the many-body electron problem in terms of $n(r)$?

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The energy functional

$$H = \sum_{i=1}^N \left[-\frac{\hbar^2 \nabla_i^2}{2m} + V_{\text{ext}}(\vec{r}_i) \right] + \sum_{i>j} \frac{e^2}{|\vec{r}_i - \vec{r}_j|}$$

$$E[n] = T[n] + E_{\text{ext}}[n] + U[n]$$

$$E_{\text{ext}}[n] = \int d^3r V_{\text{ext}}(\vec{r}) n(\vec{r})$$

$$U[n] = E_{\text{H}}[n] + \Delta U[n], \quad E_{\text{H}}[n] = \frac{e^2}{2} \int d^3r \int d^3r' \frac{n(\vec{r})n(\vec{r}')}{|\vec{r} - \vec{r}'|}$$

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Kohn-Sham: map onto a non-interacting system

- Separate, in $T[n]$, the kinetic energy of N non-interacting electrons but not necessarily homogeneous $T_s[n]$

$$T[n] = T_s[n] + T_c[n].$$

- Add the term E_{XC} that describe quantum effects (exchange and correlation) and the difference between $T[n]$ and $T_s[n]$

$$E_{xc}[n] = T_c[n] + \Delta U[n]$$

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The energy functional rewritten

$$E[n] = T[n] + E_{\text{ext}}[n] + U[n]$$

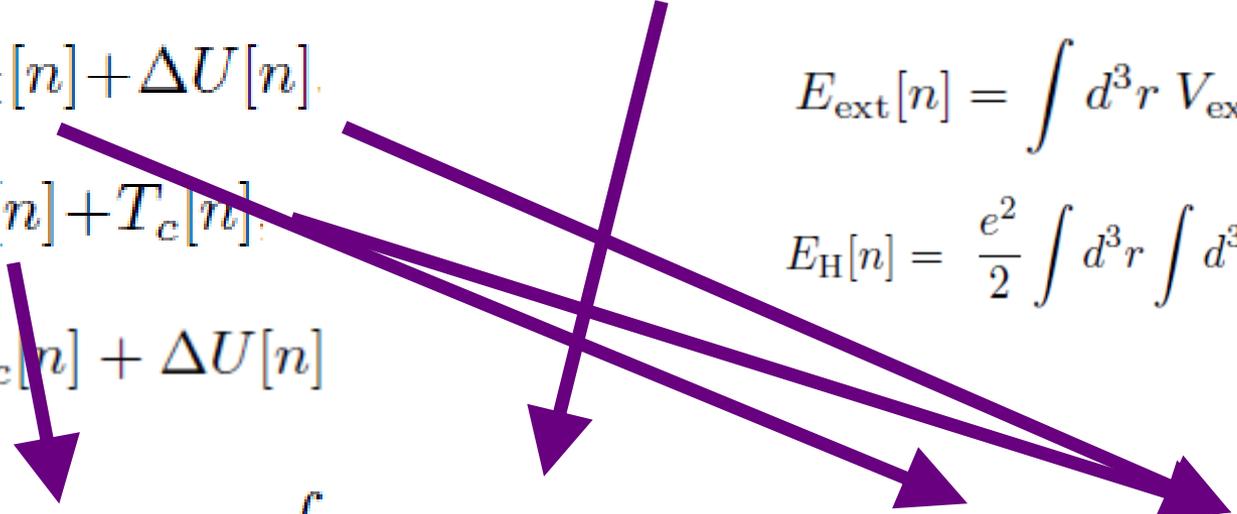
$$U[n] = E_{\text{H}}[n] + \Delta U[n]$$

$$T[n] = T_{\text{s}}[n] + T_{\text{c}}[n]$$

$$E_{\text{xc}}[n] = T_{\text{c}}[n] + \Delta U[n]$$

$$E_{\text{ext}}[n] = \int d^3r V_{\text{ext}}(\vec{r}) n(\vec{r})$$

$$E_{\text{H}}[n] = \frac{e^2}{2} \int d^3r \int d^3r' \frac{n(\vec{r})n(\vec{r}')}{|\vec{r} - \vec{r}'|}$$

$$E[n] = T_{\text{s}}[n] + \int d^3r V_{\text{ext}}(\vec{r}) n(\vec{r}) + E_{\text{H}}[n] + E_{\text{xc}}[n]$$


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The Kohn-Sham equations

- Regroup all “potential” terms into a single-particle effective potential

$$V_s(\vec{r}) = V_{\text{ext}}(\vec{r}) + e^2 \int d^3r' \frac{n(\vec{r}')}{|\vec{r} - \vec{r}'|} + V_{\text{xc}}(\vec{r}; [n])$$

$$V_{\text{xc}}(\vec{r}; [n]) = \delta E_{\text{xc}}[n] / \delta n(\vec{r})$$

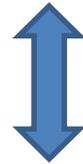
- Rewrite Schrödinger equation in terms of (solvable) independent single-particle equations

$$\left[-\frac{\hbar^2 \nabla^2}{2m} + V_s(\vec{r}) \right] \psi_j(\vec{r}) = \varepsilon_j \psi_j(\vec{r}) \quad n_s(\vec{r}) = \sum_j f_j |\psi_j(\vec{r})|^2$$

Ab initio molecular dynamics

- Solve quantum (equations, calculate ab initio enthalpies and forces (at T = 0 K)

$$E[n] = T_s[n] + \int d^3r V_{\text{ext}}(\vec{r}) n(\vec{r}) + E_H[n] + E_{\text{xc}}[n]$$



$$V(\mathbf{q}) = \frac{\langle \Psi_{\mathbf{q}} | H(\mathbf{q}) | \Psi_{\mathbf{q}} \rangle}{\langle \Psi_{\mathbf{q}} | \Psi_{\mathbf{q}} \rangle}$$

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In the end, the *ab initio* energy and forces

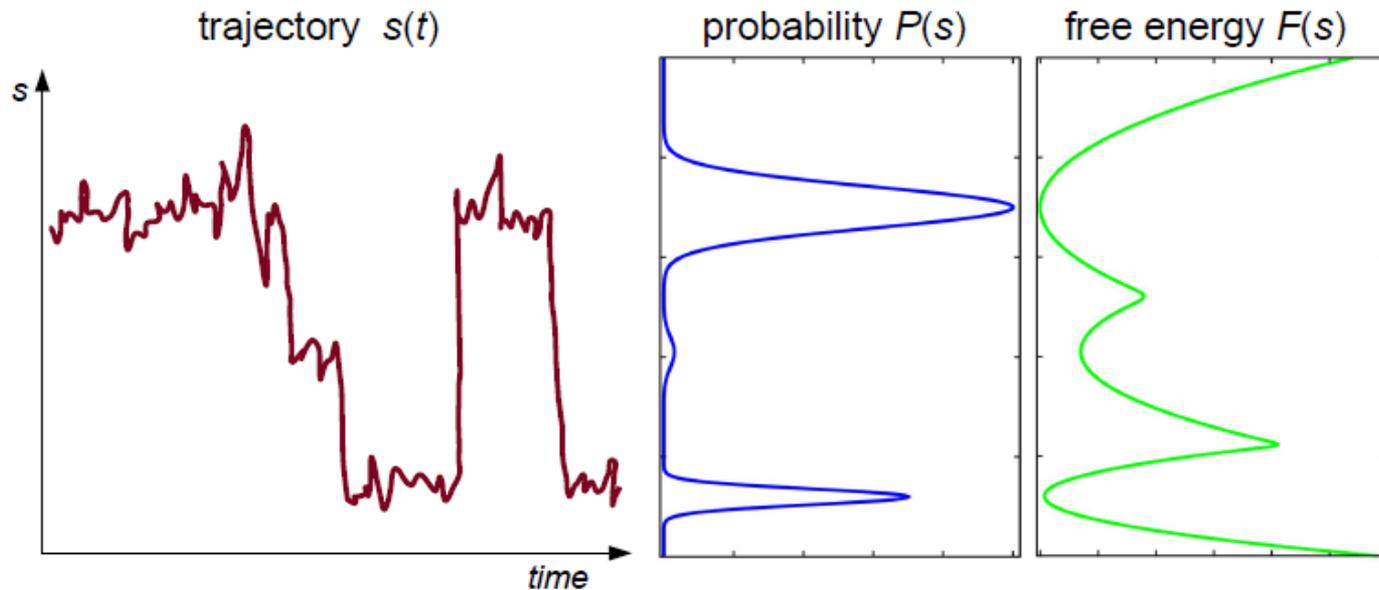
□ Solve quantum (equations, calculate ab initio enthalpies and forces (at T = 0 K)

$$E[n] = T_s[n] + \int d^3r V_{\text{ext}}(\vec{r}) n(\vec{r}) + E_H[n] + E_{\text{xc}}[n]$$

□ Finite temperature: move atoms, explore (chemical) phase space

$$\mathbf{f}_i = m \ddot{\mathbf{r}}_i \quad \mathbf{f}_i = -\frac{\partial U(\mathbf{r}^N)}{\partial \mathbf{r}_i} = -\vec{\nabla}_{\mathbf{r}_i} U(\mathbf{r}^N)$$

Connection AIMD – Statistical mechanics



from probabilities (histograms) we get free energies:

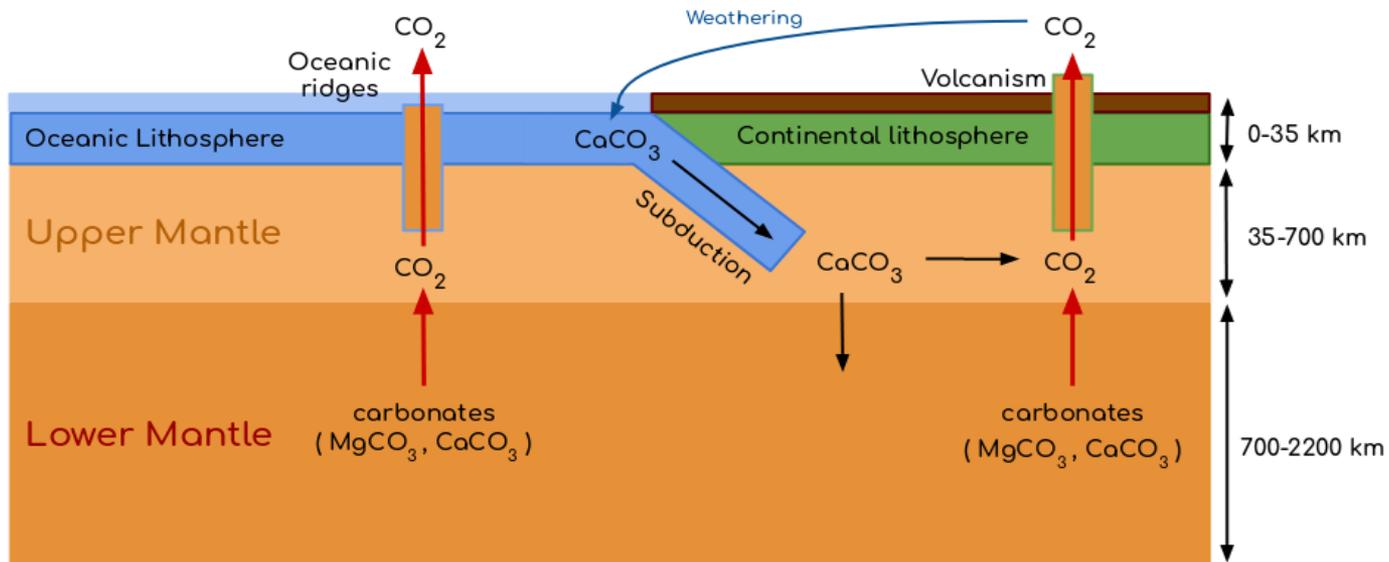
$$P(s) = \lim_{t \rightarrow \infty} \frac{1}{t} \int_0^t dt \delta(s - s(x(t))) \quad F(s) = -\frac{1}{\beta} \log P(s)$$

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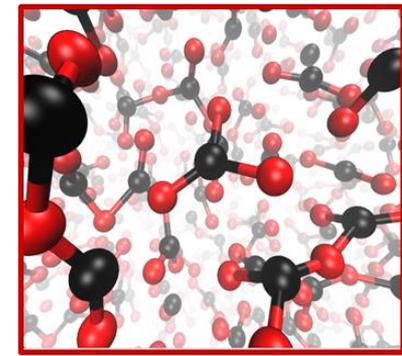
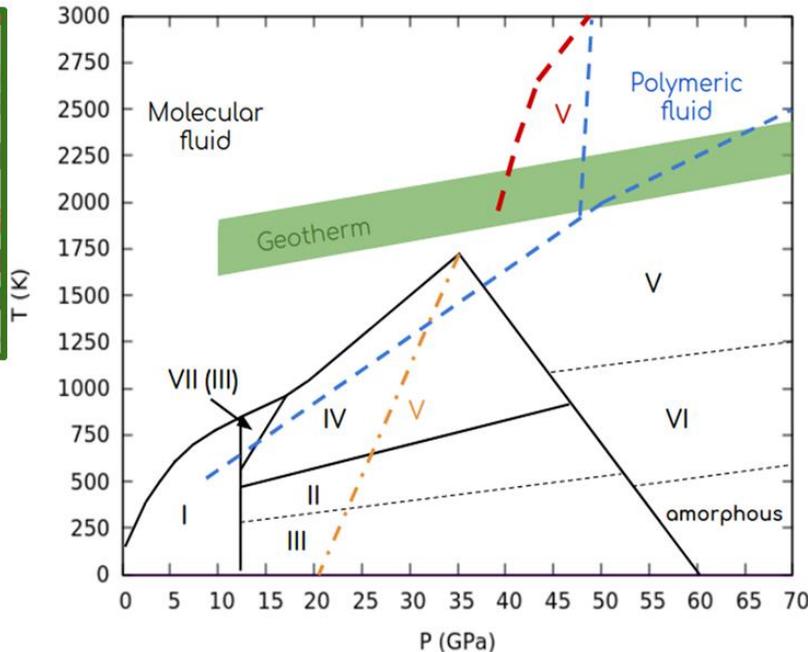
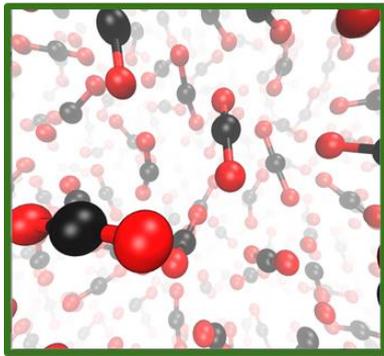
Geological fluid CO₂ (thesis of M. Moog)



- ❑ An essential component of geological physics and chemistry
- ❑ Subject to many potential “transformations”, both intrinsic and extrinsic

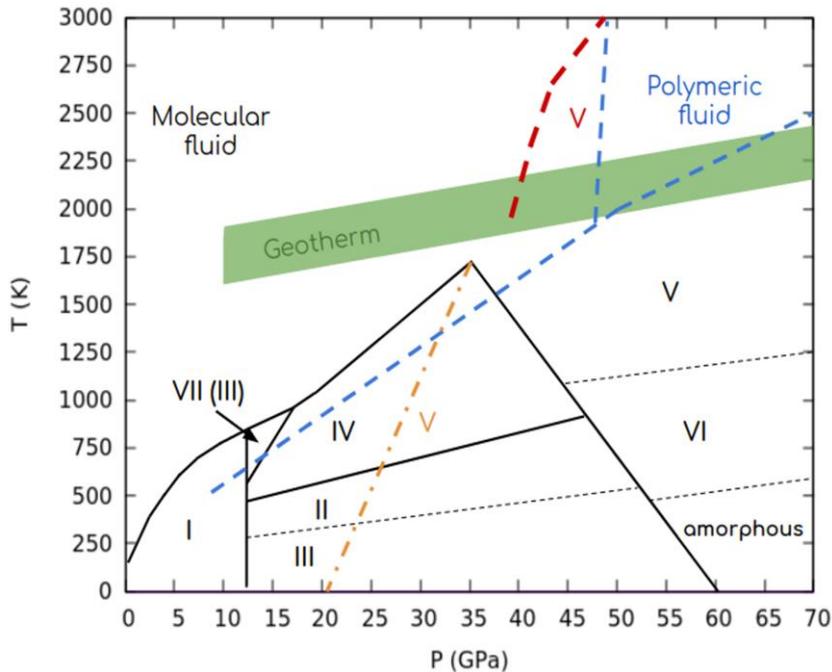
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Pressure/temperature phase diagram of CO₂



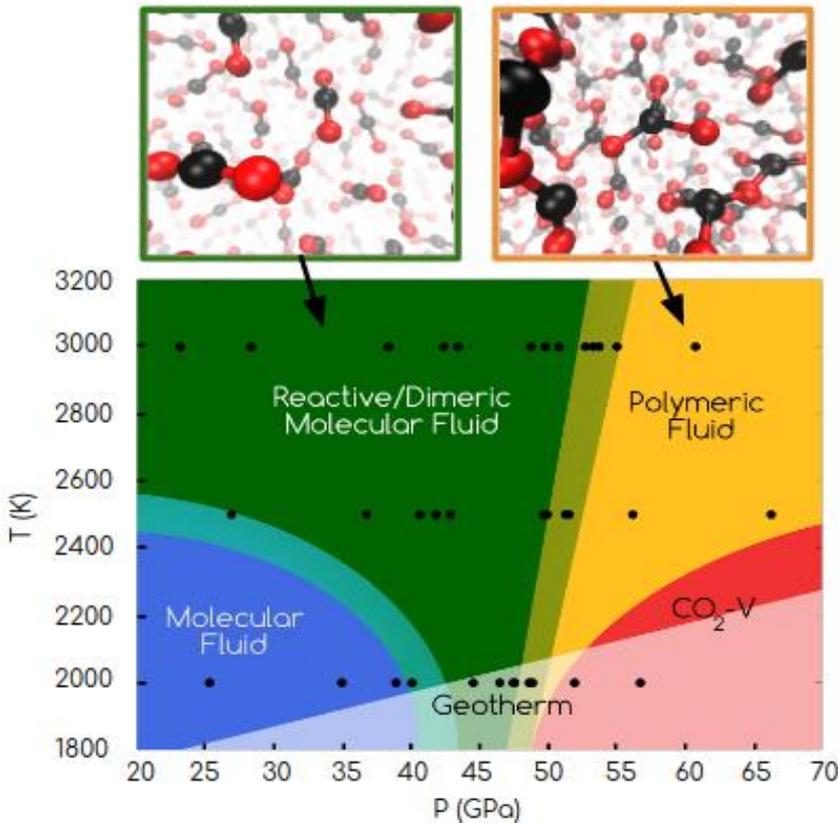
- ❑ Two major regimes: molecular (up to 40 Gpa) and polymeric (beyond)
- ❑ Many crystalline phases, and several possible disordered (liquid, amorphous) forms

CO₂ phase diagram exploration by ab initio molecular dynamics



- ❑ 45 different P/T points (3 temperatures, 15 pressures)
- ❑ 100-ps long trajectories, ~5ns trajectories in total
- ❑ 32 CO₂ molecules

Ab initio phase diagram of CO₂



Four different regimes:

- “Standard” molecular liquid: composed of individual CO₂ molecules
- Reactive/Dimeric molecular fluid: formation of dimers (and trimers), exchange of O between molecules
- Polymeric fluid: Extremely dynamic fluid with a complex chain network in perpetual evolution
- Glassy system or CO₂-V: CO₂-V likely most stable in “real” experimental conditions.

M. Moog, F. Pietrucci, AMS, *under evaluation* (2020)

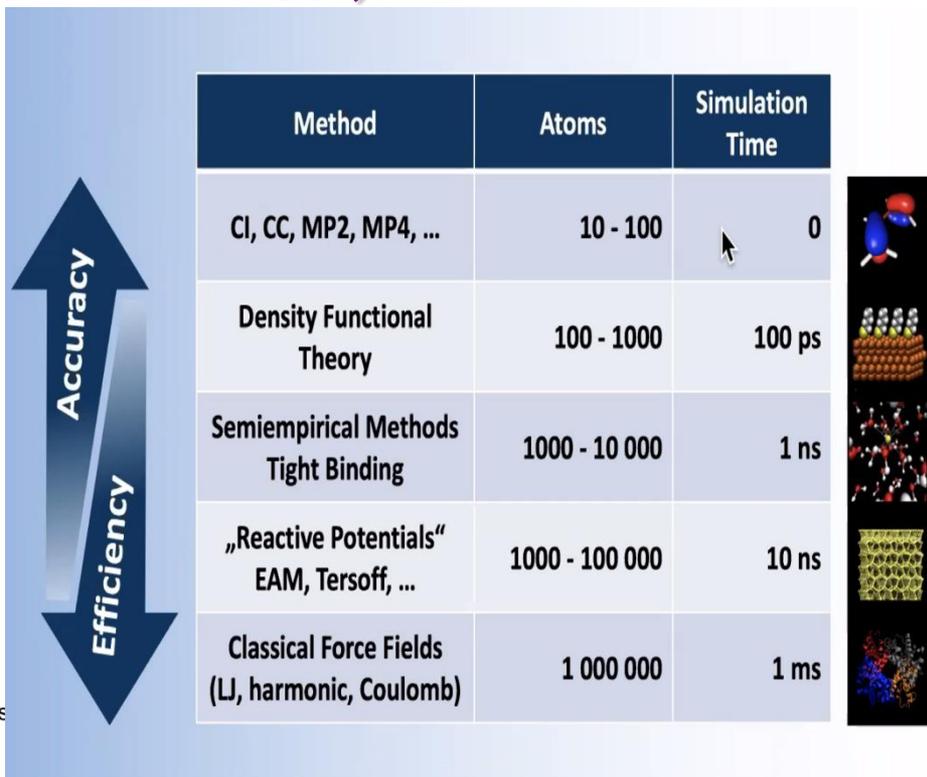
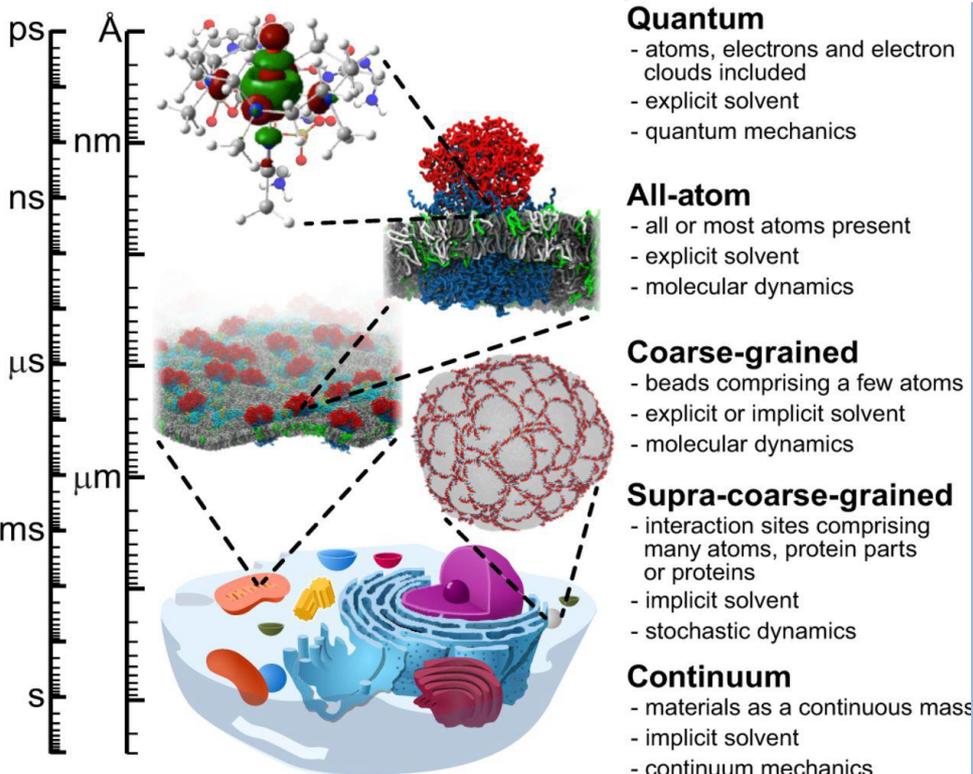
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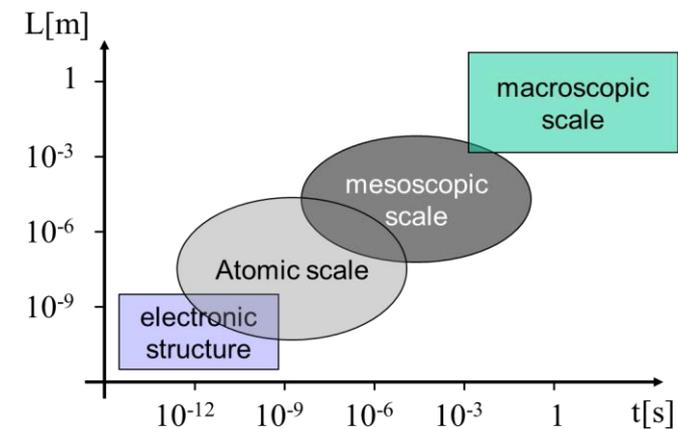


Challenge n.2 : replace costly ab initio simulations by machine-learning-based ones

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General principles of machine-learning approaches in *ab initio* computational materials science

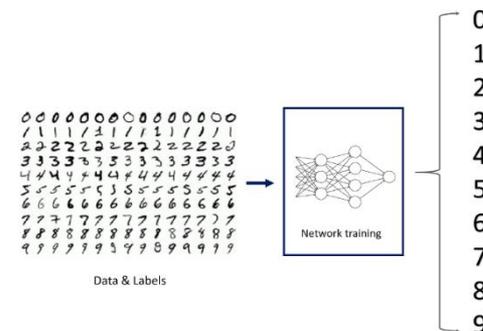
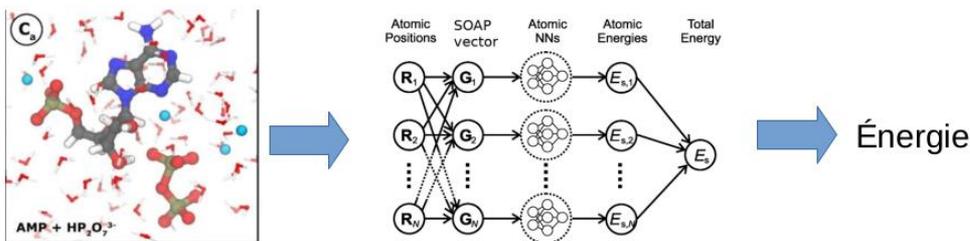
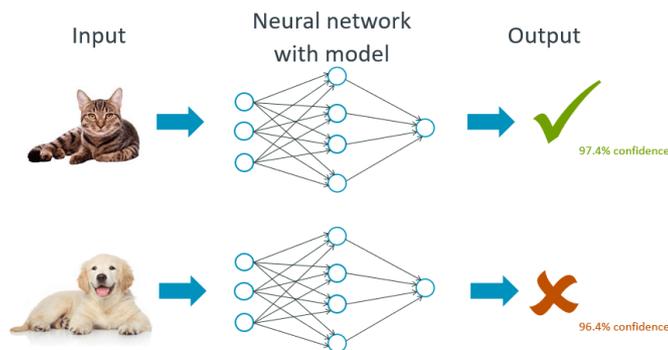
- ❑ *Ab initio* (“electronic structure”) calculations are generally very expensive
 - DFT $\sim N^3$
 - Quantum Monte Carlo $\sim N^4$
 - Adv quantum chem $\sim N^{6-7}$
- ❑ *Ab initio* trajectories $\sim ns$ timescale at the best
- ❑ CO₂ trajectories: ~ 2.5 million supercomputer hours $\sim 6-8$ months of human time at best
- ❑ Even using metadynamics or other enhanced sampling methods, this is a **major bottleneck**



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General principles of machine-learning approaches in *ab initio* computational materials science

- ❑ Neural networks can “learn”, by training, to recognize a cat from a dog, or handwritten numbers
- ❑ Can we train neural networks to “recognize” atomic configurations, and univocally associate to them physical quantities (energy, forces, etc...)?
- ❑ Can we exploit our long *ab initio* trajectories?

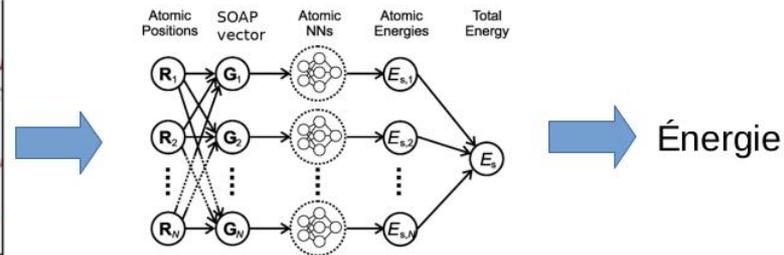
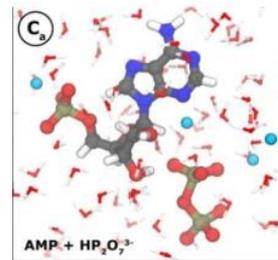


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General principles of machine-learning approaches in *ab initio* computational materials science

- ❑ “Fitting” quantum-based physical interactions is not new (it is what “classical model potentials” do)
- ❑ Founder of the method: **Jörg Behler** (Univ Göttingen)

- ❑ Building a NNP can be done:
 - without any a priori knowledge/choice of the mathematical form of the potential

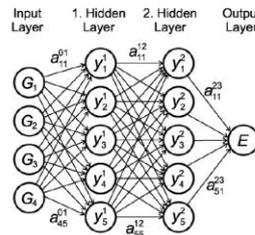
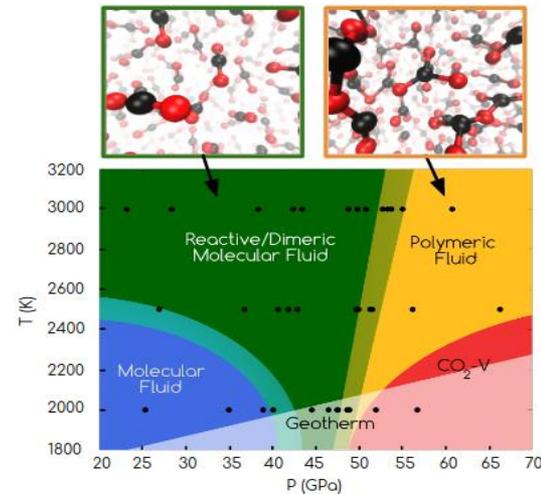


- with a “local” perspective: $E_{tot} = \sum_i^{N_{atoms}} E_i$

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- ❑ We use structural descriptors with rotational and translational symmetries: Smooth Overlap of Atomic Positions (SOAP)
- ❑ Local approach, *i.e.* for a single CO₂ molecule:
- ❑ Differences with standard fitting techniques
 - Hierarchy and non-linearities
 - Stochastic gradient descents



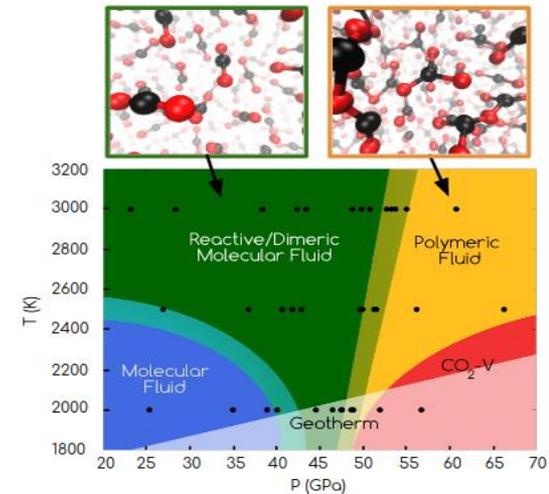
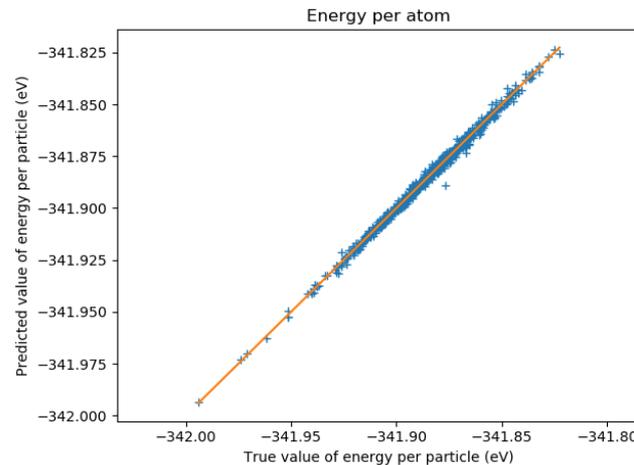
$$y_m^n = f_m^n(x_m^n) = f_m^n \left(b_m^n + \sum_{i=1}^{N_{n-1}} y_i^{n-1} a_{im}^{n-1,n} \right). \quad (3)$$

$$E = f_1^3 \left(b_1^3 + \sum_{l=1}^5 a_{1l}^{23} \cdot f_l^2 \left(b_l^2 + \sum_{k=1}^5 a_{kl}^{12} \cdot f_k^1 \left(b_k^1 + \sum_{j=1}^4 a_{jk}^{01} \cdot G_j \right) \right) \right).$$

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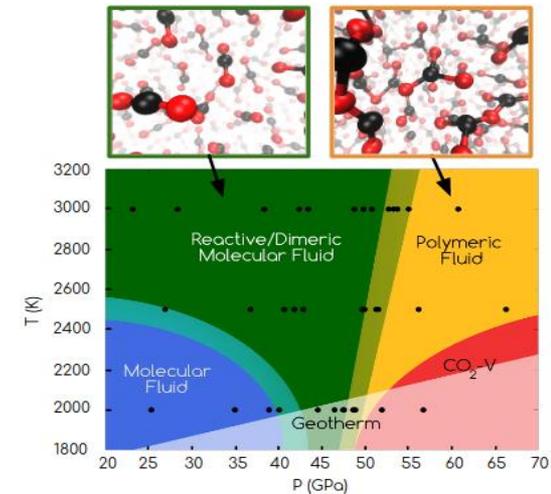
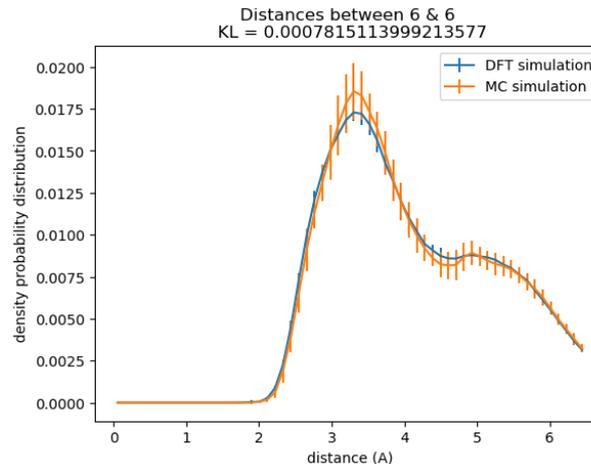
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- ❑ Evaluating the quality



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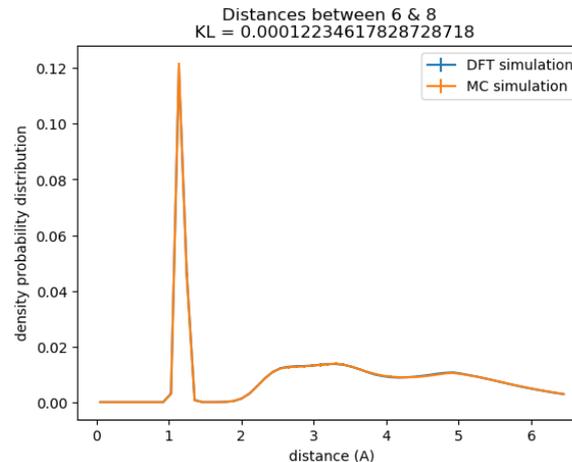
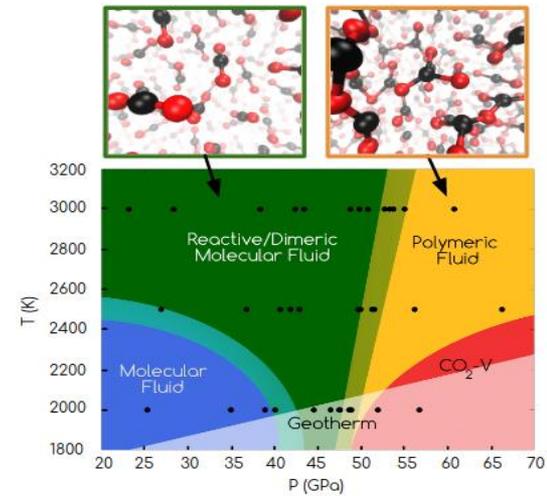
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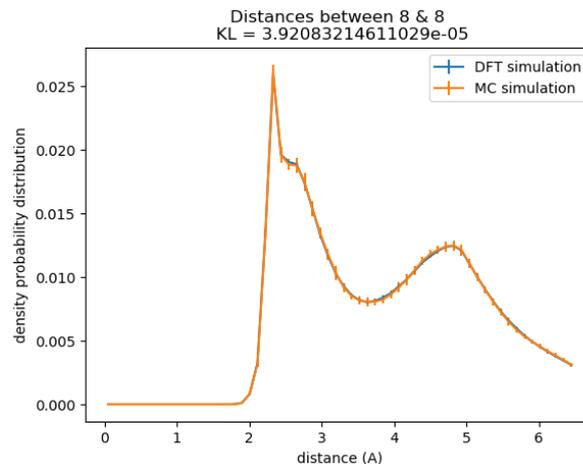
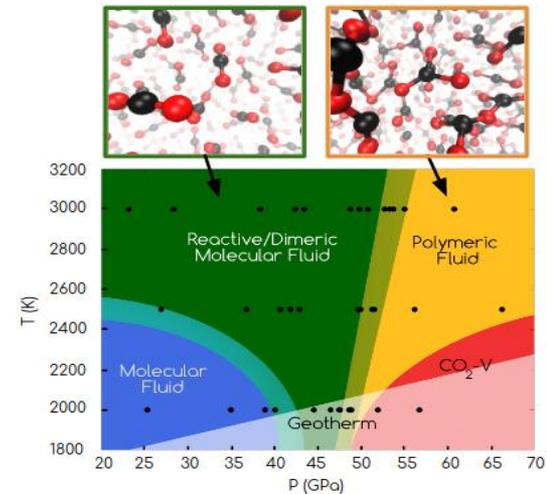
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Conclusions

- ❑ What: “generalized” structural transformations in condensed matter physics and chemistry
 - Phase transitions (under pressure)
 - Nucleation from the liquid phase
 - Chemical reactions in solutions
 - Protein structure-function relations
- ❑ **How: computer simulations of *ab initio* (= quantum) molecular dynamics**
- ❑ **One emblematic case: CO₂ at “geological” conditions**
- ❑ **Bottleneck: efficient sampling of 3N-dimensional configurational space**
- ❑ **Challenge n.1 : find “good” collective variables to describe/project the manifold onto a few relevant dimensions**
- ❑ **Challenge n.2 : replace costly *ab initio* simulations by machine-learning-based ones**

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Thanks to

**Fabio Pietrucci (IMPMC)
Michele Casula (IMPMC)
Jörg Behler (Univ Göttingen)**

**Mathieu Moog (PhD in 2019) – CO2
Julien Heu – 2nd year PhD student
Timothée Devergne – 1st year PhD student**

